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Invention: OPTICAL GLASS AND PROCESS FOR THE PRODUCTION OF OPTICAL PRODUCTS

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SPECIFICATION

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Optical Glass and Process for the Production of Optical Products

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Technical Background

[0001]

1. Field of the Invention

10 The present invention relates to an optical glass, a glass preform made of it, an optical product and a process for the production of the optical product. More specifically, it relates to an optical glass which has high-refractivity and low-dispersion optical properties, has a low glass transition point and allows the stable operation of a heat-treating furnace for a long period of time, a glass preform which is made of the above optical glass and suitable for the formation of various optical products, an optical product made of the above optical glass, and a process for the production of the above optical product.

15 [0002]

2. Explanation of Related Art

20 Conventionally, an optical glass having high-refractivity and low-dispersion optical properties contains a large amount of La_2O_3 , Gd_2O_3 , Y_2O_3 , Ta_2O_5 , ZrO_2 , etc., for attaining a high refractive index and a low dispersion property as is shown, for example, in JP-A-54-90218 and JP-B-54-6042, and glass-network-forming components such as B_2O_3 and SiO_2 are contained in a small amount, so that the above optical glass is extremely highly liable to undergo
25 crystallization. Since a glass that can be stably produced is limited in compositional range, commercially available optical glasses having a high refractive index and a low dispersion have a glass transition point T_g of over 720°C . Table 1 shows properties of high-refractivity low-dispersion
30 optical glasses shown in brochures supplied by optical glass manufacturers.

35 [0003]

Table 1

	Refractive index [nd]	Abbe's number [vd]	Glass transition point [Tg] (°C)
A	1.88300	40.8	730
B	1.88067	41.01	758
C	1.88300	40.8	738

5 [0004]

Conventional high-refractivity low-dispersion glasses have a very high temperature for viscous flows as is typically shown by glass transition points Tg, and for example, annealing treatment thereof has required a temperature of 710°C or higher. Generally, most of furnaces for glass annealing are made of a stainless steel plate, and this material has a deformation temperature around 700°C. When the annealing is carried out at a temperature over 710°C, therefore, there is caused a problem that the above stainless steel plate undergoes deformation, so that it is difficult to operate the furnace for a long period of time.

Further, the production of a lens material by re-heat pressing also requires a very high temperature, which causes a heat-treating furnace to deteriorate sooner and hinders stable production.

Meanwhile, when a glass has a glass transition point Tg of 700°C or lower, stable production has been achieved without causing any special load on the operation of facilities.

25 [0005]

Summary of the Invention

Under the circumstances, it is a first object of the present invention to provide an optical glass that has high-refractivity low-dispersion optical properties, has a low glass transition point and allows the stable operation of a heat-treating furnace for a long period of time.

It is a second object of the present invention to

provide a glass preform that is made of the above optical glass and is suitable for forming various optical products, and an optical product made of the above optical glass.

It is a third object of the present invention to
5 provide a process for efficiently producing an optical product made of the above optical glass.

[0006]

For developing optical glasses having the above desirable properties, the present inventors have made
10 diligent studies with regard to effects of compositions of components constituting glasses on optical properties, thermal properties and devitrification resistance. As a result, it has been found that an optical glass having high-refractivity and low-dispersion optical properties and having
15 a glass transition point of 700°C or lower can be obtained by controlling the ratio of the total content of La_2O_3 , Gd_2O_3 , Y_2O_3 and Yb_2O_3 , and the ratio of the total content of ZrO_2 , Ta_2O_5 and Nb_2O_5 , to the total content of SiO_2 and B_2O_3 such that the above ratios come into specific ranges, or by
20 forming a glass having a specific glass composition, and that the first object of the present invention can be accordingly achieved.

Further, it has been also found that the above second object of the present invention can be achieved by a glass
25 preform and an optical product made of the above optical glass each.

It has been further found that a optical product made of the above optical glass can be efficiently produced by employing a specific step, and that the above third object of
30 the present invention can be accordingly achieved.

[0007]

The present invention has been completed on the basis of the above findings. That is, the present invention provides:

35 (1) an optical glass having a refractive index n_d of at least 1.875, an Abbe's number v_d of at least 39.5 and a glass transition point T_g of 700°C or lower (to be referred to as "optical glass I of the present invention" hereinafter),

(2) an optical glass which is a borosilicate glass comprising at least one selected from La_2O_3 , Gd_2O_3 , Y_2O_3 or Yb_2O_3 and at least one selected from ZrO_2 , Ta_2O_5 or Nb_2O_5 , wherein the ratio (weight ratio) of the total content of La_2O_3 , Gd_2O_3 , Y_2O_3 and Yb_2O_3 to the total content of SiO_2 and B_2O_3 is from 3.2 to 5 and the ratio (weight ratio) of the total content of ZrO_2 , Ta_2O_5 and Nb_2O_5 to the total content of SiO_2 and B_2O_3 is from 1.1 to 1.5, and which has a refractive index n_d of at least 1.875 and an Abbe's number v_d of at least 39.5 (to be referred to as "optical glass II of the present invention" hereinafter),

(3) an optical glass which is a borosilicate glass comprising at least one selected from La_2O_3 , Gd_2O_3 , Y_2O_3 or Yb_2O_3 , at least one selected from ZrO_2 , Ta_2O_5 or Nb_2O_5 and ZnO , wherein the ratio (weight ratio) of the total content of La_2O_3 , Gd_2O_3 , Y_2O_3 and Yb_2O_3 to the total content of SiO_2 and B_2O_3 is from 2 to 5, the ratio (weight ratio) of the total content of ZrO_2 , Ta_2O_5 and Nb_2O_5 to the total content of SiO_2 and B_2O_3 is from 0.5 to 3 and the ratio (weight ratio) of ZnO to the total content of SiO_2 and B_2O_3 is at least 0.14, and which has a refractive index n_d of at least 1.875 and an Abbe's number v_d of at least 39.5 (to be referred to as "optical glass III of the present invention" hereinafter),

(4) an optical glass having a glass composition comprising, by % by weight, 3 to 10 % of SiO_2 , 7 to 15 % of B_2O_3 , 30 to 60 % of La_2O_3 , 2 to 8 % of ZrO_2 and 13 to 19 % of Ta_2O_5 , wherein the total content of SiO_2 and B_2O_3 is 14 to 20 % and the total content of the above components is at least 95 % (to be referred to as "optical glass IV of the present invention" hereinafter),

(5) a glass preform made of any one of the above glasses I to IV,

(6) an optical product made of any one of the above glasses I to IV,

(7) a process for the production of an optical product made of any one of the above optical glasses I to IV, which comprises the steps of melting raw materials for glass and directly press-molding a molten glass, and

(8) a process for the production of an optical product, which comprises the steps of softening a glass preform made of any one of the above optical glasses I to IV under heat and press-molding the glass preform softened under heat.

[0008]

Preferred Embodiments of the Invention

The optical glass of the present invention includes four embodiments, i.e., four optical glasses I to IV.

First, the optical glass I is a high-refractivity low-dispersion optical glass having a low glass transition point and having a refractive index n_d of at least 1.875, an Abbe's number v_d of at least 39.5 and a glass transition point T_g of 700°C or lower.

[0009]

The above optical glass I preferably includes a borosilicate glass containing at least one selected from La_2O_3 , Gd_2O_3 , Y_2O_3 or Yb_2O_3 and at least one selected from ZrO_2 , Ta_2O_5 or Nb_2O_5 , wherein the weight ratio of the total content of La_2O_3 , Gd_2O_3 , Y_2O_3 and Yb_2O_3 to the total content of SiO_2 and B_2O_3 is from 2 to 4, and the weight ratio of the total content of ZrO_2 , Ta_2O_5 and Nb_2O_5 to the total content of SiO_2 and B_2O_3 is from 1 to 2.

[0010]

In the composition of the above optical glass I, when the weight ratio of the total content of La_2O_3 , Gd_2O_3 , Y_2O_3 and Yb_2O_3 to the total content of SiO_2 and B_2O_3 [$(\text{La}_2\text{O}_3 + \text{Gd}_2\text{O}_3 + \text{Y}_2\text{O}_3 + \text{Yb}_2\text{O}_3)/(\text{SiO}_2 + \text{B}_2\text{O}_3)$] is less than 2, it is difficult to obtain an optical glass having a high refractive index and a high Abbe's number, which is one of the objects of the present invention. The content of ZnO that improves dispersion as compared with La_2O_3 and Gd_2O_3 is limited, and as a result, it is difficult to obtain an optical glass having a glass transition point sufficient for mass-productivity. On the other hand, when the above weight ratio exceeds 4, the devitrification resistance is poor, so that it is difficult to obtain a stably mass-producible glass. Therefore, the weight ratio of the total content of La_2O_3 , Gd_2O_3 , Y_2O_3 and

Yb₂O₃ to the total content of SiO₂ and B₂O₃ is preferably in the range of from 2 to 4, more preferably 3 to 4, still more preferably 3.1 to 3.7.

[0011]

Further, when the weight ratio of the total content of ZrO₂, Ta₂O₅ and Nb₂O₅ to the total content of SiO₂ and B₂O₃ $[(ZrO_2 + Ta_2O_5 + Nb_2O_5)/(SiO_2 + B_2O_3)]$ is less than 1, it is difficult to obtain an optical glass having a high refractive index intended in the present invention and stability sufficient for mass-productivity. On the other hand, when it exceeds 2, the Abbe's number ν_d decreases, and it is difficult to obtain a low-dispersion optical glass that is an end product of the present invention. Therefore, the weight ratio of the total content of ZrO₂, Ta₂O₅ and Nb₂O₅ to the total content of SiO₂ and B₂O₃ is preferably in the range of from 1 to 2, more preferably 1.1 to 1.5, still more preferably 1.2 to 1.4.

[0012]

The above optical glass I may further contain ZnO. The weight ratio of the content of ZnO to the total content of SiO₂ and B₂O₃ $[ZnO/(SiO_2 + B_2O_3)]$ is preferably more than 0 but not more than 2, more preferably more than 0 but not more than 1, particularly preferably 0.1 to 0.5. When the $[ZnO/(SiO_2 + B_2O_3)]$ weight ratio is in the above range, the glass can be imparted with properties of high refractivity and low dispersion (the dependency of a refractive index on a wavelength is small), and further, the glass can be improved in devitrification resistance and the temperature for the viscous flow thereof can be decreased.

[0013]

As a preferred composition, the optical glass I of the present invention has a glass composition (a) comprising, by % by weight, 3 to 10 % of SiO₂, 7 to 15 % of B₂O₃, 0 to 5 % of GeO₂, 0 to 15 % of ZnO, 30 to 60 % of La₂O₃, 0 to 30 % of Gd₂O₃, 0 to 10 % of Y₂O₃, 0 to 5 % of Yb₂O₃, 2 to 8 % of ZrO₂ and 13 to 19 % of Ta₂O₅, wherein the total content of SiO₂, B₂O₃ and GeO₂ is 14 to 20 % by weight, the total content of B₂O₃ and ZnO is at least 9 % by weight and the total content

of La_2O_3 , Gd_2O_3 , Y_2O_3 and Yb_2O_3 is 50 to 60 % by weight, and further wherein the total content of the above components exceeds 95 % by weight, and the composition contains 0 to 1 % by weight of Li_2O and 0 to 3 % of Nb_2O_5 .

5 [0014]

In the above glass composition, SiO_2 is a glass-network-forming component essential for maintaining devitrification resistance, and the content thereof is preferably in the range of from 3 to 10 % by weight. When
10 the above content is less than 3 % by weight, the devitrification resistance is insufficient. When it exceeds 10 % by weight, the refractive index decreases. It is therefore difficult to obtain a high-refractivity optical glass that is an end product of the present invention. In
15 view of the devitrification resistance and the refractive index, the content of SiO_2 is preferably in the range of from 6 to 9 % by weight, more preferably 6.5 to 8.5 % by weight.

[0015]

B_2O_3 is a component having an effect as a network-forming oxide or an effect on meltability of the glass and a decrease in the temperature for viscous flow, and the content thereof is preferably in the range of from 7 to 15 % by
20 weight. When the content of B_2O_3 is less than 7 % by weight, the effect on meltability of the glass and a decrease in the temperature for viscous flow is not sufficient. When it
25 exceeds 15 % by weight, the refractive index of the glass decreases. It is therefore difficult to obtain a high-refractivity glass that is an end product of the present invention. In view of the effect on the meltability of the
30 glass and a decrease in the temperature for viscous flow and the refractive index, the content of B_2O_3 is more preferably in the range of from 9 to 12 % by weight, still more preferably 9.5 to 11 % by weight.

GeO_2 has the same effect as that of the above SiO_2 ,
35 and may be incorporated in an amount in the range of from 0 to 5 % by weight. When the content thereof exceeds 5 % by weight, the devitrification resistance is liable to decrease.

[0016]

The total content of the above SiO_2 , B_2O_3 and GeO_2 is preferably in the range of from 14 to 20 % by weight. When the above total content is less than 14 % by weight, the crystallization tendency increases in intensity, so that it is difficult to obtain an optical glass that can be stably produced. When it exceeds 20 % by weight, the refractive index decreases, so that it is difficult to obtain a high-refractivity optical glass that is an end product of the present invention. In view of the crystallization tendency and the refractive index, the above total content of the above SiO_2 , B_2O_3 and GeO_2 is more preferably in the range of from 16 to 19 % by weight, still more preferably 16 to 18 % by weight.

[0017]

ZnO works to impart the glass with a high refractive index and low dispersion (the dependency of a refractive index on a wavelength is small), and further, it works to improve the glass in devitrification resistance and to decrease the temperature for the viscous flow. ZnO is therefore a component that is added as required after its amount is adjusted particularly by taking account of the amount of B_2O_3 . Specifically, $(\text{SiO}_2 + \text{B}_2\text{O}_3) : \text{ZnO} : (\text{La}_2\text{O}_3 + \text{Gd}_2\text{O}_3 + \text{Y}_2\text{O}_3 + \text{Yb}_2\text{O}_3) : (\text{Nb}_2\text{O}_5 + \text{ZrO}_2 + \text{Ta}_2\text{O}_5)$ is adjusted to impart the glass with a refractive index n_d of at least 1.875 and an Abbe's number v_d of at least 39.5 and also with devitrification resistance feasible for glass production, and the total content of B_2O_3 and ZnO is adjusted preferably to at least 9 % by weight, more preferably to at least 12 % by weight, whereby the temperature for viscous flow can be decreased (T_g is 700°C or lower). The content of ZnO is advantageously in the range of from 0 to 15 % by weight. When the content of ZnO exceeds 15 % by weight, the devitrification resistance may be insufficient in an attempt to obtain a refractive index in the intended range. On the other hand, when an attempt is made to maintain the devitrification resistance feasible for stable production, the refractive index decreases. It is therefore difficult to obtain a high-refractivity optical glass that is an end

product of the present invention. The content of ZnO is more preferably in the range of 1 to 7 % by weight, still more preferably 3 to 5 % by weight.

[0018]

5 La₂O₃ is an essential component for obtaining a high-refractivity low-dispersion optical glass, and the content thereof is preferably in the range of from 30 to 60 % by weight. When the above content is less than 30 % by weight, it is difficult to obtain an intended high-refractivity low-
10 dispersion optical glass, and when it exceeds 60 % by weight, the devitrification resistance decreases, so that it is difficult to obtain a glass that can be stably produced. The above content is more preferably in the range of from 37 to 48 % by weight, still more preferably 40 to 45 % by weight.

15 [0019]

 Gd₂O₃ may be incorporated in an amount range of from 0 to 30 % by weight as a substitute for La₂O₃. When the above content exceeds 30 % by weight, the devitrification resistance decreases, so that it is difficult to obtain a
20 glass that can be stably produced. The above content is more preferably in the range of from 0 to 18 % by weight, still more preferably 5 to 15 % by weight.

[0020]

 Y₂O₃ and Yb₂O₃ may be also incorporated in the amount
25 range of from 0 to 10 % by weight and 0 to 5 % by weight, respectively, as a substitute for the above La₂O₃. When the content of Y₂O₃ exceeds 10 % by weight, or when the content of Yb₂O₃ exceeds 5 % by weight, the devitrification resistance decreases, so that it is difficult to obtain a
30 glass that can be stably produced. The content of Y₂O₃ is more preferably in the range of from 0 to 6 % by weight, still more preferably 3 to 6 % by weight. Further, the content of Yb₂O₃ is more preferably in the range of from 0 to 5 % by weight, still more preferably 0 to 2 % by weight.

35 [0021]

 The above La₂O₃ and the above Gd₂O₃, Y₂O₃ and Yb₂O₃ have similar effects on optical properties, and preferably, the total content of these components is in the range of from 50

to 60 % by weight. When the above total content is less than 50 % by weight, it is difficult to obtain a high-refractivity low-dispersion optical glass that is an end product of the present invention. When it exceeds 60 % by weight, the devitrification resistance decreases, so that it is difficult to obtain a glass that can be stably produced. The above total content is more preferably in the range of from 51 to 58 % by weight, still more preferably 54 to 56 % by weight.

[0022]

ZrO_2 is a component for attaining a high refractive index, and it exhibits an effect on improving the devitrification resistance when incorporated in a small amount. The content thereof is preferably in the range of from 2 to 8 % by weight. When the above content is less than 2 % by weight, it is difficult to obtain a high-refractivity optical glass, and ZrO_2 may fail to exhibit the effect on fully improving the devitrification resistance. When the above content exceeds 8 % by weight, the devitrification resistance may rather decrease and the glass transition point may increase, so that it may be difficult to achieve the object of the present invention. The above content is more preferably in the range of from 4 to 8 % by weight, still more preferably 4 to 6 % by weight.

[0023]

Ta_2O_5 is an essential component for attaining the high-refractivity, and the content thereof is preferably in the range of from 13 to 19 % by weight. When the above content is less than 13 % by weight, it is difficult to obtain a high-refractivity optical glass that is an end product of the present invention. When it exceeds 19 % by weight, not only the devitrification resistance may decrease, but also the transmittance absorption end may shift toward a longer wavelength side. The above content is more preferably in the range of from 13 to 17 % by weight, still more preferably 14 to 17 % by weight.

[0024]

In the optical glass I of the present invention, preferably, the total content of SiO_2 , B_2O_3 , GeO_2 , ZnO , La_2O_3 ,

Gd₂O₃, Y₂O₃, Yb₂O₃, ZrO₂ and Ta₂O₅ exceeds 95 % by weight. When the above total content is less than 95 % by weight, it is difficult to obtain an optical glass that satisfies all of the optical properties, the decrease in temperature for viscous flow and the devitrification resistance that are intended in the present invention. The above total content is more preferably at least 96 % by weight, still more preferably at least 98 % by weight.

[0025]

In addition to the above components, the optical glass I of the present invention may contain Nb₂O₃, WO₃, Al₂O₃, Bi₂O₃, Ga₂O₃, BaO, SrO, CaO, MgO, Na₂O, K₂O, Li₂O and Sb₂O₃ as required.

[0026]

Nb₂O₃ and WO₃ are components for improving the devitrification resistance when incorporated in a small amount. Each of these components may be incorporated in an amount range of from 0 to 3 % by weight. When the content of Nb₂O₃ exceeds 3 % by weight, or when the content of WO₃ exceeds 3 % by weight, the absorption of the glass in the shorter wavelength region is intensified, which causes the glass to be colored. The content of Nb₂O₃ is more preferably in the range of from 0 to 1.5 % by weight, still more preferably 0.5 to 1.5 % by weight. Further, the content of WO₃ is more preferably in the range of from 0 to 2 % by weight, still more preferably 0 to 1 % by weight.

[0027]

Bi₂O₃ is a component having an effect on decreasing T_g when incorporated in a small amount, and it may be incorporated in an amount range of from 0 to 3 % by weight. When the content of Bi₂O₃ exceeds 3 % by weight, it decreases the devitrification resistance or causes the glass to be colored. The content thereof is more preferably in the range of from 0 to 2 % by weight, still more preferably 0 to 1 % by weight.

[0028]

Al₂O₃ and Ga₂O₃ may work to improve the devitrification resistance in some cases when incorporated in a small amount.

However, they also work to decrease the refractive index. The content of each of these components is preferably in the range of from 0 to 3 % by weight. The content of Al_2O_3 is more preferably in the range of from 0 to 2.5 % by weight, still more preferably 0 to 0.5 % by weight.

[0029]

BaO , SrO , CaO and MgO have an effect on promoting defoaming when used in the form of carbonates or nitrates as raw materials for the glass. However, when the total content of these components exceeds 3 % by weight, the devitrification resistance decreases, so that it is difficult to obtain an optical glass that can be stably produced. The total content of BaO , SrO , CaO and MgO is therefore preferably in the range of from 0 to 3 % by weight. The content of BaO is more preferably in the range of from 0 to 2 % by weight, still more preferably 0 to 1 % by weight. The content of SrO is more preferably in the range of from 0 to 2 % by weight, still more preferably 0 to 1 % by weight.

[0030]

Na_2O , K_2O and Li_2O have an effect on decreasing the glass transition point T_g , and Li_2O in particular has a very high effect on such. However, these components function as a great factor for decreasing the devitrification resistance and the refractive index. The total content of Na_2O , K_2O and Li_2O is preferably in the range of from 0 to 1 % by weight. The content of Li_2O is more preferably in the range of from 0 to 0.5 % by weight.

[0031]

Sb_2O_3 as a refining agent may be incorporated in an amount range of from 0 to 1 % by weight. The above refining agent Sb_2O_3 may be replaced with other refining agent such as SnO_2 . The content of Sb_2O_3 is preferably in the range of from 0 to 0.5 % by weight.

[0032]

As another preferred composition, the optical glass I of the present invention has a glass composition (b) comprising, by % by weight, 5 to 10 % of SiO_2 , 7 to 13 % of B_2O_3 , 0 to 5 % of GeO_2 , 0 to 15 % of ZnO , 30 to 60 % of La_2O_3 ,

0 to 30 % of Gd_2O_3 , 0 to 5 % of Y_2O_3 , 0 to 5 % of Yb_2O_3 , 2 to 8 % of ZrO_2 and 13 to 19 % of Ta_2O_5 , wherein the total content of SiO_2 , B_2O_3 and GeO_2 is 14 to 20 % by weight, the total content of B_2O_3 and ZnO is at least 9 % by weight and the total content of La_2O_3 , Gd_2O_3 , Y_2O_3 and Yb_2O_3 is 50 to 60 % by weight, and further wherein the total content of the above components exceeds 95 % by weight, the composition further contains, by % by weight, 0 to 3 % of Nb_2O_3 , 0 to 3 % of WO_3 , 0 to 3 % of Al_2O_3 , 0 to 3 % of Bi_2O_3 , 0 to 3 % of Ga_2O_3 and 0 to 1 % of Sb_2O_3 , the total content of BaO , SrO , K_2O and MgO is 0 to 3 % by weight, and the total content of Na_2O , K_2O and Li_2O is 0 to 1 % by weight.

[0033]

The optical glass II of the present invention is a borosilicate glass comprising at least one selected from La_2O_3 , Gd_2O_3 , Y_2O_3 or Yb_2O_3 and at least one selected from ZrO_2 , Ta_2O_5 or Nb_2O_5 , wherein the weight ratio of the total content of La_2O_3 , Gd_2O_3 , Y_2O_3 and Yb_2O_3 to the total content of SiO_2 and B_2O_3 is from 3.2 to 5, and the weight ratio of the total content of ZrO_2 , Ta_2O_5 and Nb_2O_5 to the total content of SiO_2 and B_2O_3 is from 1.1 to 1.5, the borosilicate glass having a refractive index n_d of at least 1.875 and an Abbe's number v_d of at least 39.5.

[0034]

The weight ratio of the total content of La_2O_3 , Gd_2O_3 , Y_2O_3 and Yb_2O_3 to the total content of SiO_2 and B_2O_3 [$(La_2O_3 + Gd_2O_3 + Y_2O_3 + Yb_2O_3)/(SiO_2 + B_2O_3)$] is from 3.2 to 5, preferably from 3.2 to 4.5, more preferably from 3.2 to 4, still more preferably from 3.2 to 3.5.

[0035]

Further, the weight ratio of the total content of ZrO_2 , Ta_2O_5 and Nb_2O_5 to the total content of SiO_2 and B_2O_3 [$(ZrO_2 + Ta_2O_5 + Nb_2O_5)/(SiO_2 + B_2O_3)$] is from 1.1 to 1.5, preferably from 1.2 to 1.3.

[0036]

While the above optical glass II has a refractive index n_d of at least 1.875 and an Abbe's number v_d of at least 39.5, it can exhibit a glass transition point T_g of

700°C or lower.

[0037]

The optical glass III of the present invention is a borosilicate glass comprising at least one selected from
5 La_2O_3 , Gd_2O_3 , Y_2O_3 or Yb_2O_3 and at least one selected from ZrO_2 ,
 Ta_2O_5 or Nb_2O_5 and containing ZnO , wherein the weight ratio of
the total content of La_2O_3 , Gd_2O_3 , Y_2O_3 and Yb_2O_3 to the total
content of SiO_2 and B_2O_3 is from 2 to 5, the weight ratio of
10 the total content of ZrO_2 , Ta_2O_5 and Nb_2O_5 to the total
content of SiO_2 and B_2O_3 is from 0.5 to 3 and the weight ratio
of ZnO to the total content of SiO_2 and B_2O_3 is at least 0.14,
the borosilicate glass having a refractive index n_d of at
least 1.875 and an Abbe's number v_d of at least 39.5.

[0038]

15 In the optical glass III, the weight ratio of ZnO to
the total content of SiO_2 and B_2O_3 [$\text{ZnO}/(\text{SiO}_2 + \text{B}_2\text{O}_3)$] is at
least 0.14, preferably from 0.14 to 2, more preferably 0.18
to 1.5, still more preferably 0.2 to 1. Further, the weight
ratio of the total content of La_2O_3 , Gd_2O_3 , Y_2O_3 and Yb_2O_3 to
20 the total content of SiO_2 and B_2O_3 [$(\text{La}_2\text{O}_3 + \text{Gd}_2\text{O}_3 + \text{Y}_2\text{O}_3 +$
 $\text{Yb}_2\text{O}_3)/(\text{SiO}_2 + \text{B}_2\text{O}_3)$] is from 2 to 5, preferably 2.5 to 4.5,
more preferably 3 to 3.5.

[0039]

Further, the weight ratio of the total content of ZrO_2 ,
25 Ta_2O_5 and Nb_2O_5 to the total content of SiO_2 and B_2O_3 [$(\text{ZrO}_2 +$
 $\text{Ta}_2\text{O}_5 + \text{Nb}_2\text{O}_5)/(\text{SiO}_2 + \text{B}_2\text{O}_3)$] is from 0.5 to 3, preferably from
0.8 to 2, more preferably from 1 to 1.5.

[0040]

While the above optical glass III has a refractive
30 index n_d of at least 1.875 and an Abbe's number v_d of at
least 39.5, it can exhibit a glass transition point T_g of
700°C or lower.

[0041]

As a preferred glass composition, the above optical
35 glasses II and III has a glass composition comprising, by %
by weight, 3 to 10 % of SiO_2 , 7 to 15 % of B_2O_3 , 0 to 5 % of
 GeO_2 , 0 to 15 % of ZnO , 30 to 60 % of La_2O_3 , 0 to 30 % of
 Gd_2O_3 , 0 to 10 % of Y_2O_3 , 0 to 5 % of Yb_2O_3 , 2 to 8 % of ZrO_2 .

and 13 to 19 % of Ta_2O_5 , wherein the total content of SiO_2 , B_2O_3 and GeO_2 is 14 to 20 % by weight, the total content of B_2O_3 and ZnO is at least 9 % by weight and the total content of La_2O_3 , Gd_2O_3 , Y_2O_3 and Yb_2O_3 is 50 to 60 % by weight, and further wherein the total content of the above components exceeds 95 % by weight and the glass composition contains 0 to 1 % by weight of Li_2O and 0 to 3 % by weight of Nb_2O_5 .

[0042]

Each component of the above glass composition is as explained with regard to the glass composition (a) in the above optical glass I.

[0043]

Further, the optical glass IV of the present invention is an optical glass having a composition comprising, by % by weight, 3 to 10 % of SiO_2 , 7 to 15 % of B_2O_3 , 30 to 60 % of La_2O_3 , 2 to 8 % of ZrO_2 and 13 to 19 % of Ta_2O_5 , wherein the total content of SiO_2 and B_2O_3 is 14 to 20 % by weight, and the total content of the above components is at least 95 % by weight.

[0044]

In the above optical glass IV, preferably, part of La_2O_3 is replaced with Gd_2O_3 and/or Y_2O_3 , the content of Gd_2O_3 is 0 to 30 % by weight, the content of Y_2O_3 is 0 to 10 % by weight, and the glass has a glass transition point T_g of 700°C or lower.

[0045]

In the optical glass IV, preferably, ZnO is contained in an amount of 0 to 15 % by weight and the total content of ZnO and B_2O_3 is at least 9 % by weight. In the optical glass IV, particularly preferably, part of La_2O_3 is replaced with Gd_2O_3 and/or Y_2O_3 , the content of Gd_2O_3 is 0 to 30 % by weight, the content of Y_2O_3 is 0 to 10 % by weight, the content of ZnO is 0 to 15 % by weight, the content of Nb_2O_5 is 0 to 3 % by weight, the content of Li_2O is 0 to 1 % by weight, and the glass has a glass transition point T_g of 700°C or lower. In the optical glass IV, a reason for limitation of the amount range of each component and a preferred amount range of each component are as explained in the glass composition (a) in

the above optical glass I.

[0046]

According to the present invention, there is also provided a preform made of any one of the above optical glasses I to IV, and there is also provided an optical product made of any one of the above optical glasses I to IV.

The above glass preform can be produced by melting raw materials for any one of the above optical glasses I to IV and pre-shaping a molten glass by cold- or hot processing.

[0047]

The above optical product can be produced by a process comprising the steps of melting raw materials for any one of the above optical glasses I to IV and directly press-molding a molten glass. Alternatively, the above optical product can be produced by a process comprising the steps of re-heating the above glass preform to a temperature suitable for press-molding (temperature to attain a viscosity of 10^4 to 10^7 Pa·s), e.g., approximately 850°C, to soften the preform, and press-molding the preform.

[0048]

The above processes may include the step of annealing the press-molded article of the glass after the step of directly press-molding a molten glass or after the step of press-molding the glass preform. The above annealing of the molded article of the glass can be carried out at a temperature around the glass transition point thereof, preferably, in a temperature range within $\pm 20^\circ\text{C}$ from the glass transition point. In the present invention, the temperature for the annealing can be set at 720°C or lower, and it can be set at a 700°C or lower for some glass compositions.

[0049]

In the above manner, optical products such as a lens and a prism can be efficiently produced.

The optical glass of the present invention is a high-refractivity low-dispersion glass having a low temperature for viscous flow as compared with a conventional optical glass. Therefore, the optical glass of the present invention

obviates any annealing or re-heat pressing at a particularly high temperature, so that the optical glass can be stably produced.

[0050]

5 The present invention will be explained more in detail with reference to Examples, while the present invention shall not be limited by any of these Examples.

[0051]

Examples 1 - 10 and Comparative Examples 1 and 2

10 Powdery raw materials of carbonate, nitrate, hydroxide, oxide, and the like were provided for forming a glass composition shown in each Example in Tables 2 to 4, and the raw materials were fully mixed. Then, the mixture was placed in a platinum crucible, melted in a furnace set at 15 1,400°C, stirred and refined, and the molten glass was cast into a frame made of iron pre-heated to a proper temperature, maintained at a temperature around T_g for 2 hours and then gradually cooled to give an optical glass.

[0052]

20 The thus-obtained optical glasses were measured for properties by the following methods. Tables 2 to 4 also show the results.

(1) Refractive index [nd] and Abbe's number [vd]

25 An optical glass was produced by cooling a molten glass at a temperature-decreasing rate of 30°C/hour and measured.

(2) Glass transition point T_g

30 An optical glass was measured with a thermo-mechanical analyzer at a temperature elevation rate of 4°C/minute.

(3) Liquidus temperature [L.T.]

35 A glass was placed in a platinum crucible having a volume of 50 ml, the crucible with a glass in it was covered and maintained in a furnace at a predetermined temperature for 2 hours and cooled. Then, an inside of the glass was observed through a microscope having a magnification of 100 times, and the liquidus temperature of the glass was determined on the basis of whether or not a crystal was

formed. The above temperature was changed at intervals of 10°C.

(4) λ_{80}

A polished sample having a thickness of 10 mm was
5 measured for a spectral transmittance, and a wavelength (nm)
at a transmittance of 80 % was determined.

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Table 2

		Example			
		1	2	3	4
Glass	SiO ₂	7.3	7.3	7.3	7.3
	B ₂ O ₃	8.4	9.4	7.6	9.4
	GeO ₂	0	0	0	0
	(SiO ₂ +B ₂ O ₃)	(15.7)	(16.7)	(14.9)	(16.7)
	(SiO ₂ +B ₂ O ₃ +GeO ₂)	(15.7)	(16.7)	(14.9)	(16.7)
	ZnO	2.2	5.0	11.0	5.0
Compo-	(B ₂ O ₃ +ZnO)	(10.6)	(14.4)	(18.6)	(14.4)
	La ₂ O ₃	47.8	55.8	41.6	45.8
	Gd ₂ O ₃	10.0	0	10.0	10.0
	Y ₂ O ₃	0	0	0	0
	Yb ₂ O ₃	0	0	0	0
	(La ₂ O ₃ +Gd ₂ O ₃ +Y ₂ O ₃ +Yb ₂ O ₃)	(57.8)	(55.8)	(51.6)	(55.8)
sition (wt%)	ZrO ₂	5.2	5.2	5.2	7.2
	Ta ₂ O ₅	14.9	15.9	15.9	13.9
	(Sub-total)	(95.8)	(98.6)	(98.6)	(98.6)
	Nb ₂ O ₅	0.8	0.8	0.8	1.2
	WO ₃	0	0	0	0
	Al ₂ O ₃	2.6	0.4	0.4	0
	Bi ₂ O ₃	0	0	0	0
	BaO	0	0	0	0
	SrO	0	0	0	0
	Li ₂ O	0.6	0	0	0
	Sb ₂ O ₃	0.2	0.2	0.2	0.2
	ZnO/(SiO ₂ +B ₂ O ₃)	0.140	0.299	0.738	0.299
	(La ₂ O ₃ +Gd ₂ O ₃ +Y ₂ O ₃ +Yb ₂ O ₃)/(SiO ₂ +B ₂ O ₃)	3.682	3.341	3.463	3.341
	(Nb ₂ O ₅ +ZrO ₂ +Ta ₂ O ₅)/(SiO ₂ +B ₂ O ₃)	1.331	1.311	1.470	1.335
Prop- erties	Liquidus temperature [L.T.] (°C)	1290	1280	1290	1260
	Refractive index [nd]	1.88	1.89	1.90	1.89
	Abbe's number [vd]	40.8	40.8	39.6	40.7
	Glass transition point [T _g] (°C)	692	699	689	697
	λ ₈₀ (nm)	471	450	467	450

Table 3

		Example			
		5	6	7	8
Glass	SiO ₂	6.7	6.7	7.3	8.3
	B ₂ O ₃	10.8	10.8	8.4	9.2
	GeO ₂	0	0	0	0
	(SiO ₂ +B ₂ O ₃)	(17.5)	(17.5)	(15.7)	(17.5)
	(SiO ₂ +B ₂ O ₃ +GeO ₂)	(17.5)	(17.5)	(15.7)	(17.5)
Compo-	ZnO	3.2	4.5	4.5	3.2
	(B ₂ O ₃ +ZnO)	(14.0)	(15.3)	(12.9)	(12.4)
	La ₂ O ₃	41.2	41.8	37.8	47.2
	Gd ₂ O ₃	10.0	9.6	18.0	10.0
	Y ₂ O ₃	6.0	3.8	0	0
sition	Yb ₂ O ₃	0	0	0	0
	(La ₂ O ₃ +Gd ₂ O ₃ +Y ₂ O ₃ +Yb ₂ O ₃)	(57.2)	(55.2)	(55.8)	(57.2)
	ZrO ₂	5.2	5.2	5.2	4.2
	Ta ₂ O ₅	15.9	15.9	13.9	16.9
	(Sub-total)	(99.0)	(98.3)	(95.1)	(99.0)
(wt%)	Nb ₂ O ₅	0.8	1.3	0	0.8
	WO ₃	0	0	1.8	0
	Al ₂ O ₃	0	0	1.9	0
	Bi ₂ O ₃	0	0	1.0	0
	BaO	0	0	0	0
	SrO	0	0	0	0
	Li ₂ O	0	0.2	0	0
	Sb ₂ O ₃	0.2	0.2	0.2	0.2
	ZnO/(SiO ₂ +B ₂ O ₃)	0.183	0.257	0.287	0.183
	(La ₂ O ₃ +Gd ₂ O ₃ +Y ₂ O ₃ +Yb ₂ O ₃)/(SiO ₂ +B ₂ O ₃)	3.269	3.154	3.554	3.269
	(Nb ₂ O ₅ +ZrO ₂ +Ta ₂ O ₅)/(SiO ₂ +B ₂ O ₃)	1.251	1.280	1.217	1.251
Prop- erties	Liquidus temperature [L.T.] (°C)	1260	1250	1290	1270
	Refractive index [nd]	1.88	1.88	1.88	1.88
	Abbe's number [vd]	41.2	40.9	40.5	41.0
	Glass transition point [T _g] (°C)	699	672	707	713
	λ ₈₀ (nm)	443	464	456	450

Table 4

		Example		CEX.	
		9	10	1	2
Glass	SiO ₂	5.9	6.7	9.3	9.8
	B ₂ O ₃	12.6	9.7	8.4	14.7
	GeO ₂	0	1.5	0.5	0
	(SiO ₂ +B ₂ O ₃)	(18.5)	(16.4)	(17.7)	(24.5)
	(SiO ₂ +B ₂ O ₃ +GeO ₂)	(18.5)	(17.9)	(18.2)	(24.5)
	ZnO	0	3.2	0	4.5
Composition (wt%)	(B ₂ O ₃ +ZnO)	(12.6)	(12.9)	(8.4)	(19.2)
	La ₂ O ₃	41.2	42.5	43.8	36.8
	Gd ₂ O ₃	11.0	10.0	14.0	7.6
	Y ₂ O ₃	5.0	4.0	0	3.8
	Yb ₂ O ₃	2.0	0	0	0
	(La ₂ O ₃ +Gd ₂ O ₃ +Y ₂ O ₃ +Yb ₂ O ₃)	(59.2)	(56.5)	(57.8)	(48.2)
	ZrO ₂	5.4	5.2	5.2	5.2
	Ta ₂ O ₅	15.9	15.9	13.9	15.9
	(Sub-total)	(99.0)	(98.7)	(96.1)	(98.3)
	Nb ₂ O ₅	0.6	0.8	0	1.3
	WO ₃	0	0	1.8	0
	Al ₂ O ₃	0	0	1.9	0
	Bi ₂ O ₃	0	0	1.0	0
	BaO	0	0.3	0	0
	SrO	0	0	0	0
	Li ₂ O	0	0	0	0.2
	Sb ₂ O ₃	0.2	0.2	0.2	0.2
	ZnO/(SiO ₂ +B ₂ O ₃)	0	0.195	0	0.184
	(La ₂ O ₃ +Gd ₂ O ₃ +Y ₂ O ₃ +Yb ₂ O ₃)/(SiO ₂ +B ₂ O ₃)	3.200	3.445	3.266	1.967
	(Nb ₂ O ₅ +ZrO ₂ +Ta ₂ O ₅)/(SiO ₂ +B ₂ O ₃)	1.184	1.335	1.079	0.914
Properties	Liquidus temperature [L.T.] (°C)	1280	1270	1290	1240
	Refractive index [nd]	1.88	1.88	1.88	1.86
	Abbe's number [vd]	41.2	41.1	40.7	42.3
	Glass transition point [T _g] (°C)	708	707	735	690
	λ ₈₀ (nm)	444	460	460	440

CEX. = Comparative Example

[0056]

As shown in Tables 2 to 4, the glasses of the present invention have a refractive index n_d of at least 1.875 and an Abbe's number v of at least 39.5, and it is seen that the glasses in Examples 1 to 6 have a glass transition point T_g of 700°C or lower and that the glasses in Examples 7 to 10 have a glass transition point T_g of 707 to 713°C.

[0057]

In the glass in Comparative Example 1, the weight ratio of $(\text{Nb}_2\text{O}_5 + \text{ZrO}_2 + \text{Ta}_2\text{O}_5)/(\text{SiO}_2 + \text{B}_2\text{O}_3)$ is 1.079 or less than 1.2, and the glass has a high glass transition point T_g of 735°C. In the glass in Comparative Example 2, the weight ratio of $(\text{La}_2\text{O}_3 + \text{Gd}_2\text{O}_3 + \text{Y}_2\text{O}_3 + \text{Yb}_2\text{O}_5)/(\text{SiO}_2 + \text{B}_2\text{O}_3)$ is 1.967 or less than 3.1, and the glass has a low refractive index of 1.86.

[0058]

When the optical glass obtained in Example 6 of the present invention was maintained in an electric furnace at 850°C for 5 minutes, it was sufficiently softened. The optical glass obtained in Comparative Example 1 was hardly softened. This difference shows that the optical glass of the present invention can be re-heat pressed at a lower temperature than a conventional optical glass.

[0059]

Example 11

Raw materials were melted in a silica crucible or a platinum crucible for forming the glass composition in each of Examples 1 to 10, and glasses were formed. Glass blocks were taken from these glasses and cut to obtain molding glass materials.

Each molding glass material was individually softened under heat until they had a viscosity of 10^4 to 10^7 Pa·s, introduced into a mold having a molding surface corresponding to an optically functional surface of an optical product as an end product, and press-molded to give glass molded materials. Then, the glass molded materials were respectively annealed at glass transition points T_g thereof, to produce optical products.

[0060]

Example 12

Raw materials were melted in a silica crucible or a platinum crucible for forming the glass composition in each of Examples 1 to 10. Each molten glass was respectively adjusted to have a viscosity of at least 0.3 Pa·s and flowed down from a feeder, and a predetermined amount of each glass that had flowed down was cast into a mold and press-molded to give glass molded materials. Then, the glass molded materials were respectively annealed at glass transition points T_g thereof, to produce optical products.

[0061]

Effect of the Invention

The optical glass of the present invention is a high-refractivity low-dispersion glass having a low temperature for viscous flow, so that it serves to attain stable production of optical products without annealing or re-heat pressing at particularly high temperatures.